

Lattice Sites of Cd in Ferroelectric BaTiO₃

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The radioactive isotope ^{111m}Cd was implanted into BaTiO₃ in order to measure electric field gradients (EFGs) with Perturbed Angular Correlation spectroscopy (PAC). It is possible to anneal the implantation induced lattice damage to a certain extent. Then, ^{111m}Cd probes are positioned at two sites with distinct axially symmetric EFGs characterized by the quadrupole coupling constants of 33.9(9) and 69.8(9) MHz. These EFGs can be assigned to ^{111m}Cd at the Ti- and the Ba-sites.

Key words: Quadrupole Interaction; Perturbed Angular Correlations; Perovskites; Nuclear Probes.

Introduction

BaTiO₃ has a perovskite structure and occurs at room temperature (RT) in the ferroelectric phase. The Electric field gradients (EFGs) at lattice sites reflect the non-cubic charge symmetry. They have been measured in detail for the Ba- and Ti-sites by NMR [1]. With Perturbed $\gamma\gamma$ -Angular Correlation spectroscopy (PAC) the EFG can be measured with the help of radioactive probe atoms that are in most cases impurities in the material under investigation. PAC investigations on BaTiO₃ have been carried out using radioactive probes that substitute for Ti: ¹⁸¹Hf(¹⁸¹Ta), ¹¹¹In(¹¹¹Cd) [2 - 4]. Recently, the influence of an external electric field on the EFG was observed [5]. These investigations shall be extended to other lattice sites than the Ti-site. In order to find a probe that substitutes Ba, we have performed experiments after implantation of ^{111m}Cd. In this report we describe the attempts to anneal the implantation induced lattice damage. Furthermore, we discuss the measured EFGs in view of the lattice sites of ^{111m}Cd in BaTiO₃.

Experimental

BaTiO₃ crystallizes in the Perovskite structure and exhibits at room temperature a tetragonal ferroelectric phase. The single crystalline sample from 'Forschungsinstitut für mineralische und metallische Werkstoffe - Edelsteine / Edelmetalle - GmbH', Idar-Oberstein, had a size of 2×2 mm² and a thickness of 1 mm. The polar axis along *c* was perpendicular to the plane. The sample had been prepared from a crystal grown with 100 ppm Co in the melt for optoelectronics use.

In perovskites, EFGs are present at lattice sites because of the non-cubic point symmetry. The experiment gives the quadrupole coupling constant ν_Q and the asymmetry parameter η of the EFG. The largest component V_{zz} of the EFG is directly proportional to the quadrupole coupling constant: $V_{zz} = \frac{h}{e} \frac{\nu_Q}{Q}$ (h = Planck's constant, e = elementary charge, Q = quadrupole moment). $\gamma\gamma$ PAC measurements have been performed with a standard 4-detector setup using BaF₂ scintillation detectors.

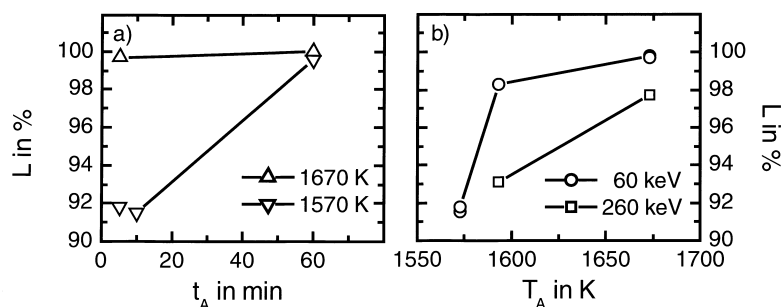


Fig. 1. Loss L of activity of BaTiO₃ during thermal treatment after the implantation of ¹¹¹mCd. a) Loss L versus annealing time t_A . Parameter is the annealing temperature of 1670 K and 1570 K. b) Loss L versus annealing temperature T_A with the implantation energy (60 keV and 260 keV) as parameter.

We have implanted ¹¹¹mCd (half live 49 min) in the single crystal at the on-line isotope separator ISOLDE at CERN with implantation energies of 60 keV as standard beam energy and a dose of $5 \cdot 10^{12} \text{ cm}^{-2}$ for each measurement. Furthermore, experiments were carried out on the high voltage platform [6] leading to a total implantation energy of 260 keV. In order to remove the implantation damage, the sample has been annealed by thermal treatment in air. It turns out that Cd tends to diffuse out of the material. As a prerequisite to PAC measurements, the annealing has been studied regarding the parameters temperature, duration of thermal treatment, time to cool down to room temperature (RT) and implantation energy. The dependence of the loss of activity, i. e. the fraction of radioactive probe atoms that leave the sample during annealing, on the annealing time is given in Fig. 1a) for two different temperatures. The temperature of $T = 1670$ K is known from the literature as the annealing temperature of BaTiO₃ implanted with ¹¹¹In [4]. Applying this procedure to BaTiO₃ implanted with ¹¹¹mCd of 60 keV leads to a loss of 100% of the activity implanted. A certain minimum temperature that leads to a partial annealing of the implantation damage is 1570 K. The loss is similar for durations of annealing of 5 and 10 min, but reaches 100% at a thermal treatment for 1 hour. The activity loss as a function of the annealing temperature is drawn in Fig. 1b) for implantation energies of 60 keV and 260 keV. At these implantation energies the ions have a range of 20 nm and 64 nm, respectively, as calculated by SRIM [7]. Implanting deeper into the crystal clearly decreases the loss of activity. The optimum procedure turned out to be the implantation with an energy of 260 keV followed by the treatment at 1570 K for 5 min and subsequent slowly pulling out the sample of the furnace within five minutes. In this way 93(2)% of the probes leave the sample and 7(2)% are available for the measurement.

In order to compare the EFG measured with ¹¹¹mCd with those measured at host element nuclei like ¹³⁷Ba, a Sternheimer correction is applied. For ¹¹¹mCd substituting Ba in BaTiO₃, the largest component of the EFG at the Ba-site is calculated as $V_{zz, \text{Cd}} = \frac{(1-\gamma_\infty)_{\text{Cd}}}{(1-\gamma_\infty)_{\text{Ba}}} V_{zz, \text{Ba}}$ ($V_{zz, X}$ = largest component of EFG observed with atom X at a certain lattice site, $(1-\gamma_\infty)_X$ = Sternheimer factor for an ion of element X [8]). The shortcomings of such considerations are well known. Nevertheless, the possible uncertainty in the quotient is much smaller than that of a single multiplier.

Results and Discussion

PAC-spectra of ¹¹¹mCd in BaTiO₃ with the corresponding Fourier transforms (time independent part subtracted prior to transformation) are displayed in Figure 2. The Fourier transform in Fig. 2a reveals two peaks that seem to be the fundamental and the 1st harmonic of one interaction. This pattern changes when the orientation of the single crystalline sample is changed. The measurement of Fig. 2b was recorded with the polar c -axis pointing 45° in between two detectors. With this orientation it is obvious that the spectrum represents two distinct EFGs at ¹¹¹mCd-sites. The EFGs have the same orientation in the crystal lattice since the Fourier amplitudes behave similarly. The parameters of the EFG are for EFG I a quadrupole coupling constant of $\nu_Q = 33.9(9)$ MHz with an asymmetry parameter of $\eta = 0.0(1)$, and for EFG II $\nu_Q = 69.8(9)$ MHz and $\eta = 0.0(3)$. 15(2)% of the probes are exposed to EFG I and the same fraction to EFG II. The remaining fraction of about 70% contributes to a broad frequency distribution caused by ¹¹¹mCd residing in highly distorted defect sites. These defects do not anneal during the thermal treatment that is described above.

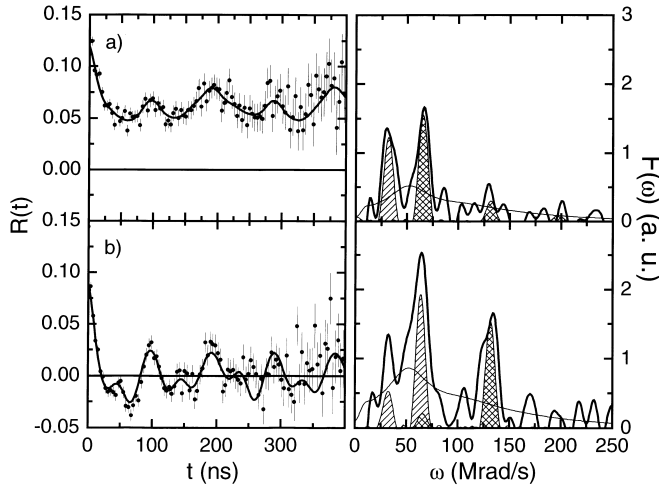


Fig. 2. PAC-spectra $R(t)$ with fit function (left) and Fourier transforms (time independent part subtracted prior to transformation) $F(\omega)$ (right) for ^{111m}Cd in BaTiO_3 measured after thermal treatment. The fit takes into account two distinct EFGs and a frequency distribution. The corresponding frequencies are marked in the Fourier transform (EFG I - striped, EFG II - checked, frequency distribution - thin line). a) Polar c -axis perpendicular to detector plane. b) Polar c -axis 45° pointing in between two detectors.

Table 1. Strength of the EFGs at the Ba- and Ti-sites in BaTiO_3 as measured with NMR [1]. Quadrupole coupling constants for ^{111m}Cd calculated from the NMR values (Sternheimer antishielding) and measured with ^{111m}Cd as well as with ^{111}In (^{111}Cd) [5].

Lattice site	NMR [1] V_{zz} in 10^{20} V/m ²	Corrected for ^{111m}Cd ν_Q in MHz	Measured ^{111m}Cd ν_Q in MHz	Measured ^{111}In (^{111}Cd) ν_Q in MHz [5]
Ba	4.35	2.8	69.8(9)	
Ti	6.82	42	33.9(9)	34.8(1)

The quadrupole coupling constant of EFG I has exactly the same value as reported in the literature for ^{111}In (^{111}Cd) substituting Ti [4, 5]. This implies that half of the ^{111m}Cd probes that occupy lattice sites with a defined EFG substitute Ti. This is surprising since Cd prefers the charge state $2+$ like Ba^{2+} , but Cd has not been reported to occur in the $4+$ state like the host ion Ti^{4+} . Since the EFG II has axial symmetry and is pointing like EFG I along the polar axis of BaTiO_3 we attribute EFG II to ^{111m}Cd at the Ba-site. The EFG at the Ba-site has not been measured with PAC but was well studied with the help of ^{137}Ba -NMR [1]. The EFGs of Ba and Ti in BaTiO_3 are given in Table 1, as well as the values for the quadrupole coupling constants that are expected for ^{111m}Cd . These values have been calculated using the Sternheimer correction. The quadrupole coupling constant for EFG II is much larger than the value expected for ^{111m}Cd at the Ba-site. This fact questions the assignment of the EFG to lattice sites.

Since the assignment of the lattice sites is ambiguous, one could imagine that EFG II is caused by a defect configuration. This does not seem to be the case. On the one hand the EFGs are reproducible for all spectra recorded. On the other hand, the orientation

of EFG II is well defined. Defects in BaTiO_3 should not have a preferred direction because after annealing the material is cooled down from the cubic phase without preferred direction to the tetragonal phase at $T = 403$ K.

Conclusions

We have demonstrated the annealing of the lattice damage after implantation of ^{111m}Cd in BaTiO_3 at ISOLDE. ^{111m}Cd is positioned at the same site as ^{111}In , which is the Ti-site [4]. A second electric field gradient is assigned to the Ba-site with some uncertainty. Emission channeling experiments might clarify the situation, although this method is rather insensitive in BaTiO_3 because of the high mass numbers of both Ba and Ti. With ab-initio calculations it will be possible in the near future to determine the relaxation around a Cd-impurity and the EFG at Cd.

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